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Key indicators

Single-crystal X-ray study T = 120 KMean σ (Mg–Zn) = 0.001 Å Disorder in main residue R factor = 0.011 wR factor = 0.029 Data-to-parameter ratio = 11.8

For details of how these key indicators were automatically derived from the article, see http://journals.iucr.org/e.

Mg and Zn disorder in Mg_{1.59(1)}Zn_{1.41(1)}Sb₂

Single crystals of magnesium zinc diantimony, $Mg_{1.59 (1)}$ -Zn_{1.41 (1)}Sb₂, were grown by a reaction of elemental Mg, Zn and Sb in an excess of Zn acting as a flux. The compound crystallizes with the trigonal *anti*-La₂O₃ type (Pearson code hP5). The structure can be regarded as a substitutional derivative of α -Mg₃Sb₂ by replacing some of the Mg atoms in the tetrahedral postion (site symmetry 3*m*.) with Zn atoms. The other two atoms in the asymmetric unit also occupy special positions, *viz*. Mg ($\overline{3m}$.) and Sb (3*m*.).

Comment

A large number of intermetallic compounds with the *anti*-La₂O₃ type (also referred to as CaAl₂Si₂ type) are known (Villars & Calvert, 1991). The difference between the two is in the cations, which in the CaAl₂Si₂ structure are on Wyckoff position 1*a*, whereas the La³⁺ ion in the La₂O₃ structure is located on one of the sites with symmetry 3*m*. (Wyckoff position 2*d*). The majority of the compounds crystallizing with



Figure 1

A projection of the crystal structure of $Mg_{1.59(1)}Zn_{1.41(1)}Sb_2$, viewed down the *b* axis. Displacement ellipsoids are drawn at the 90% probability level. Mg atoms are shown as red, the mixed Zn/Mg atoms (denoted as M^*) are shown as green, and the Sb atoms as blue ellipsoids. The unit cell is outlined.

© 2006 International Union of Crystallography All rights reserved Received 13 July 2006 Accepted 31 July 2006 this type are made up of main group elements only, although in recent years many isostructural transition metal phases have also been reported (Cordier & Schäfer, 1976; Zheng & Hoffmann, 1988; Bobev et al., 2006). By means of the extended Hückel method, the formal electron count and the apparent requirement that the CaAl₂Si₂ type is favored over the rival ThCr₂Si₂ type for d^0 , d^5 and d^{10} configurations has been explained, at least semi-quantitatively (Zheng et al., 1986; Burdett & Miller, 1990).

Fig. 1 shows a perspective view of the layered trigonal crystal structure of Mg_{1.59 (1)}Zn_{1.41 (1)}Sb₂, which can be readily described as double corrugated layers of (Zn,Mg)₂Sb₂²⁻, separated by Mg²⁺ cations. The corresponding interatomic distances are given in Table 1. The double-layers could be derived by puckering of 'dimerized' graphite-like layers or by splitting of a wurtzite-type framework and a subsequent reconstruction (Burdett & Miller, 1990). Further details on the local and extended bonding in CaAl₂Si₂-type intermetallics have been given full consideration in earlier publications (Deller & Eisenmann, 1977; Zheng et al., 1986; Bobev et al., 2006).

The structure of the title compound can be viewed as a derivative of α -Mg₃Sb₂ (Loehberg, 1934; Martinez-Ripoll *et* al., 1974), in which almost 70% of the Mg atoms occupying the tetrahedral position with Wyckoff position 2d(1/3, 2/3, z) are replaced by Zn atoms. This is supported not only by refinement of the structure with unconstrained occupancy on that site but also by the corresponding distances, which are much shorter than those observed in the related AMg₂Sb₂ compounds (A = Mg, Ca, Sr, Ba, Eu and Yb) (Deller & Eisenmann, 1977, Wartenberg et al., 2002). Analogously, the refined distances are longer than those in AZn₂Sb₂ (Mewis, 1978), in good agreement with the Zn and Mg covalent radii (1.364 Å for Mg and 1.213 Å for Zn; Pauling, 1960). These experimental findings are well supported by the local bonding and site preference analyses (Zheng et al., 1986; Burdett & Miller, 1990) based on band structure calculations.

Lastly, we point out that similar disorder of Mg and Zn has been proposed for the isostructural solid solution Mg_{1.86}Zn_{1.14}Sb₂ (Loehberg, 1934); however, this report errorneously suggests that Mg on both the octahedral and tetrahedral positions is replaced by Zn. Our refinement of Mg_{1.59 (1)}Zn_{1.41 (1)}Sb₂, which is the Zn-richer phase, unambiguously shows that the octahedral Mg position [Wyckoff symmetry 1a, (0,0,0) is not disordered.

Experimental

Handling of the pure metals and of the reaction mixtures was carried out either in an argon-filled glove-box or under vacuum. All starting materials were used as received: Mg (Alfa, turnings, 99.98% metal basis), Zn (Alfa, shot, 99.999%), Sb (Alfa, ingot, 99.99%). The synthesis was performed by loading the elements in a ratio Mg:Zn:Sb = 1:10:2 into an alumina crucible, which was subsequently sealed in an evacuated fused silica jacket. The reaction mixture was heated to 1273 K at a rate of 300 K h⁻¹, allowed to equilibrate at this temperature for 5 h, and cooled slowly to 873 K at a rate of 5 K h^{-1} . After reaching this temperature, it was removed from the furnace and the excess of molten Zn was removed by centrifugation. Black, airstable crystals of the title compound were the only product of this reaction.

Crystal data

$Mg_{1.59}Zn_{1.41}Sb_2$	$D_x = 5.180 \text{ Mg m}^{-3}$
$M_r = 374.32$	Mo $K\alpha$ radiation
Trigonal, P3m1	$\mu = 18.11 \text{ mm}^{-1}$
a = 4.3966 (9) Å	T = 120 (2) K
c = 7.171 (3) Å	Irregular fragment, black
V = 120.04 (6) Å ³	$0.08 \times 0.06 \times 0.04 \text{ mm}$
Z = 1	
Data collection	
Bruker SMART APEX	1294 measured reflections
diffractometer	118 independent reflections
ω scans	117 reflections with $I > 2\sigma(I)$
Absorption correction: multi-scan	$R_{\rm int} = 0.023$
(SADABS; Sheldrick, 2003)	$\theta_{\rm max} = 26.1^{\circ}$
$T_{\min} = 0.287, T_{\max} = 0.482$	

Refinement

Refinement on F^2	$w = 1/[\sigma^2(F_o^2) + (0.017P)^2]$
$R[F^2 > 2\sigma(F^2)] = 0.011$	where $P = (F_0^2 + 2F_c^2)/3$
$wR(F^2) = 0.029$	$(\Delta/\sigma)_{\rm max} < 0.001$
S = 1.29	$\Delta \rho_{\rm max} = 0.37 \ {\rm e} \ {\rm \AA}^{-3}$
118 reflections	$\Delta \rho_{\rm min} = -0.69 \text{ e } \text{\AA}^{-3}$
10 parameters	

Table 1

Selected bond lengths (Å).

Mg—Sb	3.0403 (6)	$\frac{Zn1/Mg1\!-\!Sb}{Zn1/Mg1\!-\!Zn1/Mg1^{ii}}$	2.7740 (13)
Zn1/Mg1—Sb ⁱ	2.7470 (6)		3.0685 (10)
		1. (;;)	

Symmetry codes: (i) -x + 1, -y + 2, -z + 1; (ii) -x, -y + 1, -z + 1.

In the structure refinement, the full occupancies for all sites were verified by freeing the site occupation factor for an individual atom, while the remaining occupation parameters were kept fixed. This proved that the Sb site (site symmetry 3m.) and one of the Mg sites (site symmetry $\overline{3}m$) are fully occupied with deviations from full occupancy within 3σ . The second Mg site (site symmetry 3m.) refined with occupancy much higher than 100%. When Zn was refined in that position, the occupancy factor was significantly below unity. Therefore, this position (labeled Zn1/Mg1) was modeled as a statistical mixture of Zn and Mg, and refined as 70.3 (4)% Zn and 29.7 (4)% Mg.

Data collection: SMART (Bruker, 2002); cell refinement: SAINT (Bruker, 2002); data reduction: SAINT; program(s) used to solve structure: SHELXTL (Sheldrick, 2001); program(s) used to refine structure: SHELXTL; molecular graphics: XP in SHELXTL; software used to prepare material for publication: SHELXTL.

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